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Hydrogen Sensing Apparatus and Method

Field of Invention

The present invention relates to apparatus and a method for measuring the concentration of hydrogen in fluid media at elevated temperatures using a high temperature protonconducting solid electrolyte in conjunction with an internal hydrogen standard.

Background of the Invention

The monitoring and control of hydrogen concentration in gaseous and liquid media is an important technological 10 issue. Fields of application include, for instance, the analysis of gas composition on the fuel side of hydrogen-based fuel cells and the determination of dissolved hydrogen content in molten metals like aluminium. It is therefore desirable to develop simple, 15 easily applicable, reliable and inexpensive sensors having high sensitivity and selectivity.

One concept of constructing hydrogen sensors for operation at elevated temperatures is to utilise a proton conducting solid electrolyte that compares the hydrogen partial pressure on the measuring side with a known and fixed hydrogen partial pressure on the reference side. appropriate proton conducting solid electrolytes are perovskites, with doped strontium cerate (SrCe_{0.95}Yb_{0.05}O_{3-d}) and doped calcium zirconate (CaZr_{0.9}In_{0.1}O_{3-d}) being applied most frequently. Under the relevant experimental conditions, these materials exhibit predominant proton conductance. Electrodes are formed by covering the surface of the electrolyte with a catalytically active and electronically conducting material, for instance platinum. If two electrodes on different areas of the same

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electrolyte body are brought into contact with two media of different hydrogen contents, i.e., p'H2 and p"H2, a hydrogen concentration cell is formed:

p'H2. | proton conducting solid electrolyte | p"H2

The potential difference generated may be described in terms of the well known Nernst equation, where U is the electromotive force (emf), R is the universal gas constant, T is the absolute temperature, F is Faraday's constant, and p"H2 and p'H2 are the hydrogen partial pressures at the measuring electrode and the reference electrode, respectively:

$$U = -\frac{RT}{2F} \ln \frac{P''_{H_2}}{P'_{H_2}}$$

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Measuring the potential difference between the two electrodes and knowing the hydrogen partial pressure on the reference side, yields the unknown hydrogen partial pressure on the measuring side.

The incorporation of a hydrogen reference standard into the sensor unit constitutes a scientific and technological problem. Two different types of hydrogen sensors employing a solid electrolyte in conjunction with a hydrogen reference have thus far been reported.

The most straightforward approach consists in the utilisation of a gaseous hydrogen standard [T. Yajima, K. Koide, N. Fukatsu, T. Ohashi and H. Iwahara, Sensors and Actuators B 13-14, 697 (1993); T. Yajima, K. Koide, H. Takai, N. Fukatsu and H. Iwahara, Solid State Ionics 79, 333 (1995)]. This requires a cell design in which one side of the solid electrolyte is in contact with the medium to be analysed while the other side is continuously supplied with a reference gas mixture of known hydrogen partial pressure. A hydrogen analyser for use in molten

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aluminium, based on this principle and using $CaZr_{0.9}In_{0.1}O_{3-d}$ as the solid electrolyte, has been developed and commercialised. However, the use of a reference gas has been found awkward, and no breakthrough with this technology has been achieved.

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In alternative approaches attempts have been made to fix the hydrogen partial pressure on the reference side by means of solid compounds or mixtures of solid compounds. The utilisation of hydrates like Ce₂(SO₄)₃•8H₂Q and AlPO4.0.34 H2O as the reference in conjunction with 10 $SrCe_{0.95}Yb_{0.05}O_{3-d}$ and $CaZr_{0.9}In_{0.1}O_{3-d}$ as the solid electrolyte has been reported [H. Iwahara, H. Uchida, T. Nagano and K. Koide, Denki Kagaku 57, 992 (1989); T. Yajima, K. Koide, K. Yamamoto and H. Iwahara, Denki Kagaku 58, 547 (1990); T. Yajima, H. Iwahara, K. Koide and K. Yamamoto, 15 Sensors and Actuators B 5, 145 (1991)]. However, incorporation of hydrates fixes the water rather than the hydrogen partial pressure and, even though some response behaviour to hydrogen has been observed in a few cases, these sensors require calibration and their signal 20 stability is insufficient for practical applications. utilisation of a calcium/calcium hydride (Ca/CaH2) mixture as the reference in contact with $SrCe_{0.95}Yb_{0.05}O_{3-\delta}$ as the solid electrolyte has been reported [M. Zheng and X. Zhen, Solid State Ionics 59, 167 (1993); M. Zheng and X. Zhen, 25 Met. Trans. B 24, 789 (1993); M. Zheng and X. Chen, Solid State Ionics 70/71, 595 (1994)]. However, this combination was only found to work at comparatively low temperatures, i.e., below 600°C, and for relatively short times, i.e., a few hours, otherwise a continuous drift of 30 the sensor signal towards zero was observed. The reason for the failure was identified to be the chemical instability of the electrolyte/reference interface. causes a chemical reaction between the highly reducing reference material and the oxide-based solid electrolyte, 35 which gradually converts the ion (proton) conductor into a

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mixed conductor and renders sensor readings impossible to interpret. Overall, no hydrogen sensor relying on a solid reference material has as yet proven to be viable in any practical application.

5 Summary of Invention

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The invention provides apparatus and methods for sensing hydrogen concentration as defined in the appended independent claims. Preferred or advantageous features of the invention are set out in dependent subclaims.

The present invention may thus provide an apparatus for measuring hydrogen concentration, comprising a proton-conducting solid electrolyte in conjunction with a self-contained and hermetically sealed metal/hydrogen reference standard, of which the content and/or the spatial distribution of oxygen is appropriate substantially to prevent chemical reaction between the solid electrolyte and the reference material, particularly at the interface there between.

The present invention may thus advantageously provide a sensor with a novel hydrogen standard that establishes a defined and reproducible reference hydrogen partial pressure and ensures chemical stability of the electrolyte/reference interface.

The present invention is based on the realisation that, first, a metal/hydrogen two-phase/two-component mixture (being a solution of hydrogen in the metal such that, under the conditions of use of the apparatus, the solution lies within a two-phase field of the metal-hydrogen phase diagram) may be used as an internal hydrogen standard in sensors employing an oxide-based proton-conducting solid electrolyte, because this type of mixture is able to fix the hydrogen partial pressure inside an encapsulated

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volume adjacent to the electrolyte and, second, may advantageously enable the interface between the electrolyte and the reference to be chemically stable. is further realised that, in a preferred embodiment, the second issue may be fulfilled by maintaining a suitable oxygen activity in the reference material, which is both sufficiently high in order to guarantee chemical stability in contact with the oxide-based electrolyte, so the proton conducting properties of the latter are not affected, and sufficiently low in order not to invalidate the two-phase/two-component approach. It should be noted that the appropriate oxygen concentration, or range of oxygen concentration, required to achieve this in any particular case may depend not only on the oxygen activity required for proper operation of the reference standard but also on the chemical stability of the electrolyte material.

According to one embodiment of the invention, the proton conducting solid electrolyte is a perovskite, preferably $SrCe_{0.95}Yb_{0.05}O_{3-\delta}$ or $CaZr_{0.9}In_{0.1}O_{3-\delta}$, and the metal component of the metal/hydrogen reference system is titanium, zirconium or hafnium. For these materials the above requirements may readily be met, as will be pointed out in the following.

The metal in the reference standard may be an alloy and the reference standard may contain other elements which affect its phase diagram. Nevertheless, the quantitative predominance of the respective metal and hydrogen in the two-phase mixture guarantee that the chemical potential and the activity of the two components, i.e., the respective metal and hydrogen, are thermodynamically fixed in terms of Gibbs' phase rule. This means that, within the range of the two-phase area (within which the two phases of the metal can coexist), the hydrogen activity is independent of the composition of the reference system and also does not change when the composition undergoes small

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variations during sensor operation. The hydrogen activity of the reference system may be determined from literature data and is only a function of temperature. Knowledge of the reference hydrogen partial pressure for the given temperature permits direct determination of the hydrogen partial pressure on the measuring side. In case of titanium, the α -titanium/ β -titanium two-phase region is preferred whilst the β -titanium/ δ -titanium two-phase region is less suitable because the corresponding hydrogen partial pressures are beyond atmospheric pressure. Regarding zirconium, both the α -zirconium/ β -zirconium and the β -zirconium/ δ -zirconium two-phase areas may be used, but the latter is preferred because of its extended composition range at elevated temperatures. Concerning hafnium, only the α -hafnium/ δ -hafnium two-phase region is appropriate.

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Secondly, a chemically stable interface between the solid electrolyte and the reference material may advantageously be ensured. It is important to note that even minute changes in the oxygen concentration may have a dramatic impact on the electrochemical properties of oxide-based proton conducting solid electrolytes. In fact, the release of small amounts of oxygen has been shown to convert these materials from pure proton conductors into mixed conductors, oxygen ion conductors or semiconductors, which makes them inappropriate for the application envisaged. Accordingly, very reactive metals like alkali metals, alkaline earth metals and rare earth metals, which also form two-phase areas with hydrogen, are not preferred for use as the reference material, since they reduce the oxide-based solid electrolyte at elevated temperatures. Even less reactive metals like titanium, zirconium and hafnium may, in their pure state, be sufficiently reducing to affect the performance of the solid electrolyte. However, and in contrast to the previously mentioned metals, the reactivity of titanium, zirconium and hafnium

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may be lowered considerably through the presence of only small amounts of oxygen. In this way, the electrolyte/reference interface may be rendered chemically stable, whilst the two-component/two-phase approach is not compromised.

The signal of a sensor, which is constructed in accordance with the above requirements, may advantageously be used to determine directly the hydrogen content in a fluid medium. If the composition of the medium needs to be controlled, the composition may then be varied until the required signal is recorded.

Description of the Drawings

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Figure 1 is a schematic illustration of an apparatus according to an embodiment of the invention;

- Figure 2 is a plot of the measured cell potential when using sensors with the α-titanium/β-titanium + hydrogen (+ oxygen) reference system to measure hydrogen concentration in hydrogen/argon gas mixtures of known hydrogen concentration at different temperatures;
- Figure 3 is a plot of the measured cell potential when using sensors with the β -zirconium/ δ -zirconium + hydrogen (+ oxygen) reference system to measure hydrogen concentration in hydrogen/argon gas mixtures of known hydrogen concentration at different temperatures;
- Figure 4 is a plot of the measured cell potential when using sensors with the α -hafnium/ δ -hafnium + hydrogen (+ oxygen) reference system to measure hydrogen concentration in hydrogen/argon gas mixtures of known hydrogen concentration at different temperatures; and

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Figure 5 is a schematic illustration of an apparatus according to a second embodiment of the invention.

Description of the Invention

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Figure 1 shows a schematic illustration of a preferred embodiment of the invention, comprising a solid electrolyte body 1, a reference material 2, an inert packing material 5, a glass seal 6, a catalytic coating at a reference electrode 3, a catalytic coating at a measuring electrode 4, a lead to the reference electrode 7, a lead to the measuring electrode 8, and an electronic measuring unit 9.

The solid electrolyte is shaped as a tube, closed at one end, with a length of about 20mm and a diameter of about 5mm, but it may be appreciated that the precise dimensions are not critical. This solid electrolyte shape may be described as a thimble. In the preferred embodiment, the electrolyte material is a perovskite. A catalytic coating may be applied to the interior and the exterior surfaces of the electrolyte tube. Electrical leads may be placed on both surfaces. In the preferred embodiment, the catalytic coatings and the electrical leads are made from platinum.

Typically, about 50 to 200mg of the reference material may be used, but it may be appreciated that the exact quantity is not critical. Lower amounts as small as 10mg or 15mg may also be used, but 70mg to 90mg or about 80mg has been found to be most effective. In the preferred embodiment, the reference material is titanium/hydrogen, zirconium/hydrogen or hafnium/hydrogen and is placed inside the electrolyte tube.

The reference material is encapsulated by means of a.. suitable sealing material. When applying an oxide-based

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sealing glass, the silicon content must be low in order to prevent detrimental reactions between the hydrogen in the reference compartment and the silicon in the glass, which would result in decomposition of the reference material. In the preferred embodiment, a silicon-free glass based on the oxides of aluminium, barium, boron, calcium and magnesium is used. The direct contact of the reference material and the sealing material may be detrimental. In the preferred embodiment, an inert packing material like pure calcium zirconate or yttrium oxide serves as a separator between both these components.

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It may be appreciated that other designs of the invention may likewise be employed. These may include layered designs, in which use is made of pellets or films, which may be printed. In these, the solid electrolyte body, the reference material and the inert packing material (if required) are used in a parallel arrangement, such as in a stack of layers. This arrangement or stack is sealed, such that only the measuring electrode on the electrolyte body is exposed to the ambient medium.

An example is shown in Figure 5, in which an electrolyte layer 20 is placed beneath a reference standard layer 22, both formed as disc-shaped pellets. A packing material 24 covers the upper and side surfaces of the reference standard layer and the stack thus formed is sealed in a glass casing 26, leaving only one face of the electrolyte exposed for access to media in which hydrogen concentration is to be measured. The packing material separates the reference standard layer from the sealing glass to prevent chemical degradation. Electrical connections to the probe are formed by layers applied to the upper and lower electrolyte surfaces, in the same way as described in other embodiments.

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Preparation of the apparatus is straightforward and can be performed in two ways. The first procedure consists of two steps. In the first step, a quantity of titanium, zirconium or hafnium metal is inserted into the open end of the solid electrolyte tube, or thimble, and a seal across the open end of the tube is created by melting and then solidifying a solder glass under an atmosphere of an inert gas or hydrogen gas or a mixture thereof. residual oxygen content should be low in order to avoid oxidation of the metal. The seal ensures that the metal is in contact with the electrolyte but hermetically sealed from the environment. In the second step, and depending on the amount of hydrogen present in the reference compartment after sealing, an electric current is applied such that hydrogen is electrochemically transported into or out of the reference compartment until the metal to hydrogen atomic ratio is suitable for the metal/hydrogen mixture to function as a reference standard for hydrogen. This method of preparation is preferred for the use of titanium/hydrogen as the reference system.

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The second procedure consists of only one step. a quantity of titanium, zirconium or hafnium metal is inserted into the open end of the solid electrolyte tube, or thimble, and a seal is created by melting and solidifying a solder glass under a hydrogen-containing atmosphere while, simultaneously, the reference is being formed through hydrogen uptake by the metal from the gas. In order for the metal/hydrogen mixture to function as a standard for hydrogen, it is important to match the melting temperature of the glass and the hydrogen content of the gas atmosphere such that, after formation of the seal, the metal to hydrogen atomic ratio in the metal/hydrogen reference is inside the desired two-phase area. This method of preparation is preferred for the use of zirconium/hydrogen or hafnium/hydrogen as the reference system.

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After preparation of the apparatus according to one of the above procedures and prior to use, preconditioning is carried out at elevated temperature, preferentially beyond 700°C, in a humidified gas atmosphere of low hydrogen partial pressure, preferentially below 1% by volume.

The apparatus may be placed directly into the medium to be analysed, which may be stagnant or flowing, at a temperature sufficient for the solid electrolyte to conduct ionically. Preferably the temperature is in the range of 500°C to 900°C. The sensors were found to detect hydrogen contents from at least 100ppm to 100% by volume.

Examples

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High density ceramic thimbles of indium oxide doped calcium zirconate ($CaZr_{0.9}In_{0.1}O_{3-d}$) were obtained through isostatic pressing of a suitable powder and sintering at 1600°C in air for 8h. Porous platinum electrodes were generated by firing a platinum-containing ink at 1000°C in air for 1h. Platinum lead wires were attached to both platinum coatings.

20 Example 1

40mg of titanium metal pieces, cut from a grit-blasted sheet of commercial grade 4 titanium metal with a known bulk oxygen content of 3600ppm by mass, were placed inside a ceramic calcium zirconate thimble. (Grit-blasting was carried out to clean the surfaces of the as-received titanium metal specimen.) The interior of the thimble was then filled with undoped calcium zirconate powder which is inert and acts as a packing material. This was covered with a layer of a laboratory-made, silicon-free, sealing glass powder, which has a melting point of approximately 930°C. To melt the glass and form the seal, the arrangement was heated to around 940°C in an alumina tube

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under pure hydrogen. Prior to application, the hydrogen was passed through calcium sulphate to remove traces of moisture and through a suitable metal scrubber to ensure low residual oxygen content. The unit was then exposed to a 1% by volume hydrogen in argon gas mixture at 700°C and coulometric titration was performed. To that end, a direct current of around 60mA, this typically corresponding to voltages in the range of a few hundred millivolts, was applied for about 200h, with the inner electrode connected to the positive terminal and the outer electrode connected to the negative terminal. By way of this procedure, a quantity of hydrogen was removed from the reference compartment, such that the titanium to hydrogen ratio established in the reference system was inside the α -titanium/ β -titanium two-phase area. After preparation, the sensor was preconditioned at 800°C in argon, which had been humidified by passing through a water bubbler at room temperature, for at least 1h. Sensor measurements were performed between 500 and 800°C in hydrogen/argon mixtures with hydrogen contents of 10ppm, 100ppm, 1%, 10% and 100% by volume. Measured émfs are shown in Figure 2. The data are in good agreement with thermodynamically expected values. Sensor signals were stable, with a drift of typically less than 1mV/d, and the response time to changes in temperature and hydrogen partial pressure was in the order of minutes. Variations in the results for different sensors were found to be less than 5%.

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Notably, although this good performance was obtained with grade 4 titanium, no stable sensor readings were obtained when pieces of grit-blasted grade 1 or grade 2 titanium metal sheets with bulk oxygen contents of 1450 and 1780 ppm by mass, respectively, were applied as the metal component in the reference system. This suggests the importance of the oxygen content in the reference system for proper sensor performance.

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This observation that the indium oxide doped calcium zirconate electrolyte was reduced by grade 1 or grade 2 titanium but not by grade 4 titanium suggests an acceptable range of oxygen concentration for this combination of materials. However, different electrolyte materials used with titanium-based reference standards may require different oxygen concentrations in the titanium. For example, a more stable electrolyte may tolerate lower oxygen concentrations in the titanium.

10 Example 2.

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About 100mg of zirconium metal were cut from a commercial zirconium wire with a known bulk oxygen content of 1500ppm by mass and placed inside a ceramic calcium zirconate thimble. The interior of the thimble was filled with yttrium oxide powder, which acts as an inert packing 15 material, and this was covered with a layer of siliconfree sealing glass powder as described in example 1. melt the glass and form the seal, the arrangement was. heated to around 940°C in an alumina tube under pure hydrogen. By way of this procedure, a zirconium to 20 hydrogen ratio inside the β -zirconium/ δ -zirconium two-phase area was established directly. Preconditioning of the sensor was carried out as described in example 1. Sensor measurements were performed between 500°C and 800°C in hydrogen/argon mixtures with hydrogen contents of 1, 10 25 and 100% by volume. Measured emfs are shown in Figure 3. Agreement with thermodynamic expectations, signal stability and comparability between individual sensors were even better than what was found, and reported in example 1, for sensors relying on the titanium/hydrogen 30 reference system.

Notably, the above zirconium material could be employed successfully both in the as-received and in the grit-blasted state. In contrast, a different zirconium

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wire, with a bulk oxygen content of 1010ppm by mass, was found to work successfully only in the as-received state, then providing similar results to the ones shown in Figure 3. When applying the same zirconium material after grit-blasting, no stable signals were achieved. This suggests that the particular material possesses an oxygen-rich surface layer which renders the electrolyte/reference interface stable if used in the as-received state, but that the bulk oxygen content is too low to allow for a stable interface once the outer layer is removed.

Example 3 -

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About 200mg of hafnium metal were cut from a commercial hafnium wire with a known oxygen content of 230ppm by mass and placed inside a ceramic calcium zirconate thimble. 1.0mg of titanium dioxide was added. The interior of the thimble was filled with yttrium oxide powder, which acts as an inert packing material, and this was covered with a layer of a laboratory-made silicon-free sealing glass powder, which has a melting point of approximately 970°C. To melt the glass and form the seal, the arrangement was heated to around 980°C in an alumina tube under pure hydrogen. By way of this procedure, a hafnium to hydrogen ratio inside the α -hafnium/ δ -hafnium two-phase area was established directly. Preconditioning of the sensor was carried out as described in example 1. measurements were performed between 600 and 800°C in hydrogen/argon mixtures with hydrogen contents of 1, 10 and 100% by volume. Measured emfs are shown in Figure 4. Sensor performance was again found to be good.

Notably, the above hafnium wire could be used neither in the as-received nor in the grit-blasted state. This suggests that, firstly, the bulk oxygen content is too low to allow for a stable electrolyte/reference interface and,

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secondly, that the oxygen-rich surface layer, if any, is too thin to prevent oxygen uptake of the reference material from the electrolyte. So, it is only through the formation of a passivating surface layer, brought about by the decomposition of titanium dioxide in the presence of hydrogen gas and subsequent precipitation of oxygen-containing species on the hafnium wire, that stability of the electrolyte/reference interface is provided.